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PHASE TRANSITIONS AND “NONMETALLIC” TEMPERATURE DEPENDENCE OF CONDUCTION ELECTRON SPIN RESONANCE LINE WIDTH IN QUASI-TWO-DIMENSIONAL SYNTHETIC METAL $C_{15}HNO_3$

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In the synthetic metal $C_{15}HNO_3$ during and after crystallization of the HNO_3 “guest” molecules, the simultaneous increase of conductivity, conduction electron spin resonance (CESR) line width and spin carriers concentration (N) was found and investigated. The increase of N is directly connected with the partial localization of the conduction π -electrons (in terms of graphite intercalation compounds tight binding model). Since the localized electrons are the perturbation centres for the conduction electrons, it has been proposed that the increase of their concentration is the main reason for the decrease of the in-plane spin carriers mobility and the “nonmetallic” increase of the CESR line width observed during and after crystallization of the HNO_3 “guest” layers.

Keywords: A. metals, D. electronic transport, D. phase transitions, E. electron paramagnetic resonance.

1. INTRODUCTION

THE TEMPERATURE dependence of the conduction electron spin resonance (CESR) line width (ΔB) in metals has been successfully explained [1] in terms of the Elliot’s theory [2] according to which $\Delta B = \alpha(\Delta g)^2/\tau$, where α is a constant, Δg is the g -shift from the free electron value, and τ is the conductivity relaxation time. The frequency of the electron–electron and electron–phonon collisions decreases with temperature and, as a result, τ increases, while the concentration (N) of the current carriers remains practically constant because of large values of Fermi temperature [3]. As a result, when the temperature lowers, the conductivity of pure metals $\sigma = Ne^2\tau/m^* = Ne\mu$ (m^* , e and μ are the effective mass, charge and mobility of charge carriers, respectively) increases, and simultaneously, if Δg is temperature independent, the CESR line width decreases.

In graphite intercalation compounds (GICs) of the acceptor type $C_{15}HNO_3$ we found a relation between σ and ΔB other than in pure metals. In this synthetic metal during and after intercalate crystallization Δg is

unchanged, while both the electroconductivity and the CESR line width increase simultaneously. In this paper we report the results of an investigation of the CESR line width “nonmetallic” temperature dependence origin in GIC $C_{15}HNO_3$.

2. EXPERIMENTAL

GICs comprise a wide class of synthetic metals and consist of an alternating sequence of n hexagonal graphite monolayers (n is the stage index) and a monolayer of “guest” atoms or molecules (intercalate) [4]. The compounds $C_{15}HNO_3$ investigated belong to the third stage of α -modification of GICs with nitric acid of the general formula $C_{5n}HNO_3$ ($n = 1, 2, 3, \dots$) [4, 5]. According to the data of various physical methods [4], in these GICs the two-dimensional liquid-like layers of HNO_3 are ordered and form a two-dimensional crystal at temperatures lower than $T \sim 250$ K. Layers of HNO_3 may be incommensurate with a carbon net along one of its crystallographic directions and they undergo a structural phase transition of the incommensurate phase–commensurate phase type at $T \sim 210$ K [6].

CESR spectra of GIC $C_{15}HNO_3$ plates were registered at X-band ($\nu = 9.52$ GHz) in the temperature range 100–300 K using the rectangular resonator with the TE_{102} mode and 2.5 kHz modulation of an external constant magnetic field B_0 . At the conventional setting of the rectangular resonator, the electrical component of the microwave field is parallel to B_0 ; in the geometric center of the resonator the magnetic component of the microwave field is parallel to the vertical axis of the resonator.

Highly oriented pyrolytic graphite plates required for GIC synthesis were cut out of a single bar with the basal plane conductivity equal to $(1.2 \pm 0.2) \times 10^4 \text{ Ohm}^{-1} \text{ cm}^{-1}$. These plates had a shape of a rectangular parallelepiped with dimensions: width (l) \times height (h) \times thickness (d), where $l \times h$ is the square of a basal plane. Accuracy in determining the size of plates was $\sim 5 \times 10^{-4} \text{ cm}$. Synthesis of GIC $C_{15}HNO_3$ was carried out in nitric acid with the density $\rho = 1.49 \text{ g cm}^{-3}$. The GIC stage was controlled by the diffraction method. The effect of temperature on CESR line shape parameters was investigated using the plate of size $0.40 \times 0.40 \times 0.02 \text{ cm}^3$.

3. RESULTS AND DISCUSSION

In the entire temperature range of investigations a single asymmetric CESR signal of $C_{15}HNO_3$ is observed. Down to ~ 247 K the asymmetry parameter of the first derivative of the absorption line A/B (which is defined as the ratio of the maximum peak height (A) to the minimum peak height (B), both measured with respect to the zero line of the resonance derivative) and the width of the CESR line at half height of the peak A are independent of temperature and are equal to 3.3 ± 0.1 and $6.0 \times 10^{-5} \text{ T}$, respectively. At all temperatures, the spectrum is axial with respect to the c -axis and is characterized by $g_{\parallel} = 2.0023 \pm 0.0001$ and $g_{\perp} = 2.0028 \pm 0.0001$. The line width increases as temperature decreases below 247 K (Figs 1 and 2). Simultaneously, the A/B value decreases, and it is equal to 2.9 ± 0.1 at 110 K (Fig. 2). The derivative $d(\Delta B)/dT$ has peaks at $T_{c1} \approx 244 \text{ K}$, $T_{c2} \approx 242 \text{ K}$ and $T_{c3} \approx 212 \text{ K}$ (Figs 1 and 2). As the temperature increases, ΔB and A/B values change inversely but with a "global" temperature hysteresis (Figs 1 and 2). Temperatures T_{c1}^- (T_{c1}^+) and T_{c2}^- (T_{c2}^+) are close to the temperature of the intercalate crystallization (melting) in $C_{15}HNO_3$ ($T_c \sim 250 \text{ K}$) known from literature [4]. This allows us to treat the CESR line shape transformations observed in $C_{15}HNO_3$ at T_{c1}^- (T_{c1}^+) and T_{c2}^- (T_{c2}^+) as a result of the two-step change in the aggregate

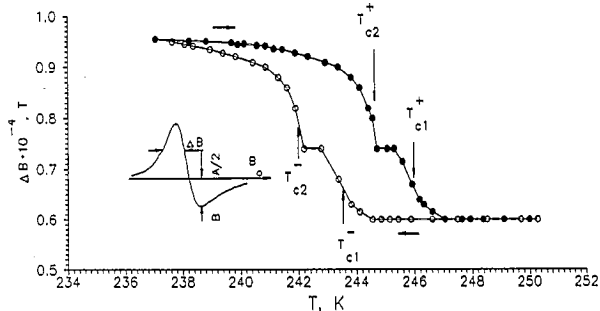


Fig. 1. Temperature dependence of CESR linewidth ΔB of $C_{15}HNO_3$ plate with dimensions: $0.4 \times 0.4 \times 0.02 \text{ cm}^3$ during crystallization (unfilled symbols) and melting (filled symbols) of intercalate. T_{c1}^- (T_{c1}^+) and T_{c2}^- (T_{c2}^+) are the temperatures of the first and second peaks of the derivative $d(\Delta B)/dT$ at cooling (heating) of the sample, respectively. $B_0 \perp c$, $\nu = 9.52$ GHz.

state of the "guest" molecules. The CESR line shape transformation at T_{c3} , is probably due to the incommensurate–commensurate phase transition. At powers of the microwave field far from saturation of the CESR signal and at the same temperature, ΔB values in Q - and X -bands coincide indicating that the CESR line is homogeneously broadened.

The basal plane electroconductivity (σ_a) of $C_{15}HNO_3$ plates was measured by the contactless induction method using a device analogous to that

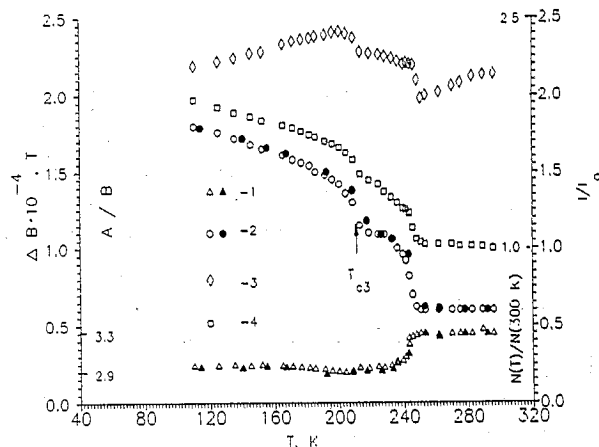


Fig. 2. Temperature dependence of CESR line parameters of $C_{15}HNO_3$ plate with dimensions: $0.4 \times 0.4 \times 0.02 \text{ cm}^3$, i.e., $A/B(T)$ (curve 1), $\Delta B(T)$ (curve 2), $I(T)/I_0$ (curve 3), where $I = (A + B)\Delta B^2$ and I_0 is the intensity of the Mn^{2+} ESR line in a standard sample $ZnS:Mn^{2+}$, and the relative spin concentration $N(T)/N(300 \text{ K})$ (curve 4), determined by the expression (1). Unfilled (filled) symbols correspond to experimental values of parameters at cooling (heating) of the GIC. T_{c3} is the temperature of the third peak of the derivative $d(\Delta B)/dT$. $B_0 \perp c$, $\nu = 9.52$ GHz.

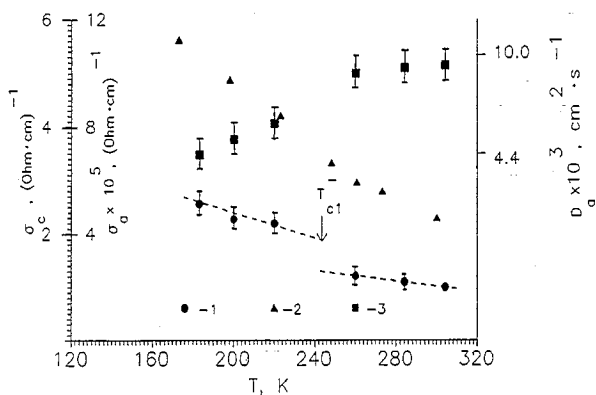


Fig. 3. Temperature dependence of $\sigma_c(1)$, $\sigma_a(2)$ and $D_a(3)$ in $C_{15}HNO_3$. The dashed line corresponds to the linear function $\sigma_c = [(-4.77 \times 10^{-3} \text{ K}^{-1}) T + 2.45] (\Omega \text{ cm})^{-1}$ and $\sigma_c = [(-1.76 \times 10^{-2} \text{ K}^{-1}) T + 5.79] (\Omega \text{ cm})^{-1}$ for $T > T_{c1}^-$ and $T < T_{c1}^-$, respectively.

described by Pendry *et al.* [7]. The results of σ_a measurements are presented in Fig. 3, and indicate that the σ_a increases as temperature decreases.

In the microwave field of a given configuration only the conduction plate regions adjacent to the vertical faces ($h \times l$) and ($h \times d$), situated approximately within the skin-depth, contribute to the CESR [8, 9]. In $C_{15}HNO_3$ one may neglect the contribution to CESR from regions adjacent to the basal planes ($h \times l$) due to the high conduction anisotropy ($\sim 10^5$) [4]. This peculiarity of $C_{15}HNO_3$ plates allows us to analyze their CESR line shape, in particular, at contactless determination of the c-axis electroconductivity (σ_c) (see below), using the one-dimensional Dyson expression [10] for CESR line shape in isotropic metals and well known Feher and Kip [11] or Kodera [12] nomograms calculated on the basis of this expression.

The σ_c measurement is not a trivial problem. The contactless method to determine the σ_c of acceptor GICs was suggested by Saint-Jean and McRae [13]. They recommended the use of the well-known A/B vs l/δ_c nomograms [8, 9, 12, 13] at interval $l/\delta_c < 2.5$ for this purpose. At this interval the ratio of A/B does not depend on spin carrier mobility [8, 9, 12, 13]. This enables one to determine the value of δ_c and, consequently, the value of σ_c unambiguously, by measuring the values of A/B and l . However, in narrow samples of GICs with HNO_3 [14] as well as in narrow samples of GICs with AsF_5 [13] the values of σ_c and spin carrier diffusion constant differ from values of this parameter in wide samples. This unusual result for narrow GIC samples is likely connected with the fact that the intercalate layers' composition and organization depend on a sample size [14]. That is why another contactless method was used to determine the value of

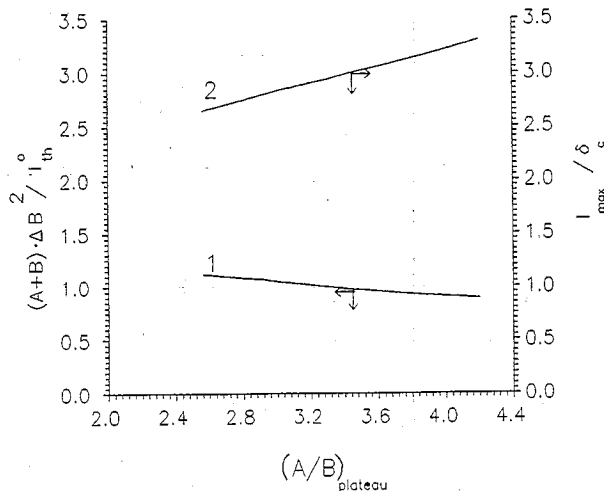


Fig. 4. The relation of theoretical value of intensity $(A+B)\Delta B^2/I_{th}^0$ (curve 1) and coordinate of the maximum (l_{\max}/δ_c) of the theoretical dependence $A/B(l/\delta_c)$ (curve 2) plotted against the value of A/B on the "plateau" of the $A/B(l/\delta)$ dependence (at $l/\delta_c > 10$). I_{th}^0 is the value of the I_{th} at $A/B = 3.3$ (in quasi-liquid phase of the intercalate).

σ_c in our study. The principles of this method are as follows.

It follows from the one-dimensional Dyson expression for CESR line shape in metals [10] that the coordinate of the maximum (l_{\max}/δ_c) of the $A/B(l/\delta_c)$ dependence and the value of asymmetry parameter on "plateau" (at $l/\delta_c > 10$) of this dependence [$(A/B)_{\text{plat}}$] are unambiguously related to each other. The corresponding nomogram calculated by authors is presented in Fig. 4. The theoretical value of l_{\max}/δ_c can be found from this nomogram, if the experimental value of $(A/B)_{\text{plat}}$ is known. Then, at a given frequency of the microwave field, the value of σ_c can be easily calculated, if the experimental value of l_{\max} is known as well. It must be emphasized that the given procedure of σ_c -determination is applicable only in the case when (1) one may neglect the contribution to CESR from regions adjacent to the basal planes ($h \times l$) of GIC plates and (2) the CESR line shapes of plates with $l \simeq l_{\max}$ as well as plates with l from "plateau" of the $A/B(l)$ dependence should be described by the one-dimensional Dyson expression for CESR line shape [10] with the same value of σ_c and T_{Da} . Because of this, before the procedure under consideration was applied to determine the σ_c -value in $C_{15}HNO_3$, the experimental $A/B(l)$ dependence was studied in more detail at fixed temperature both before (at ~ 300 K) and after (at 220 and 183 K) the phase transition. Then this dependence was approximated by the corresponding theoretical curve calculated from the one-dimensional Dyson expression for CESR line shape [10] with the value of σ_c determined

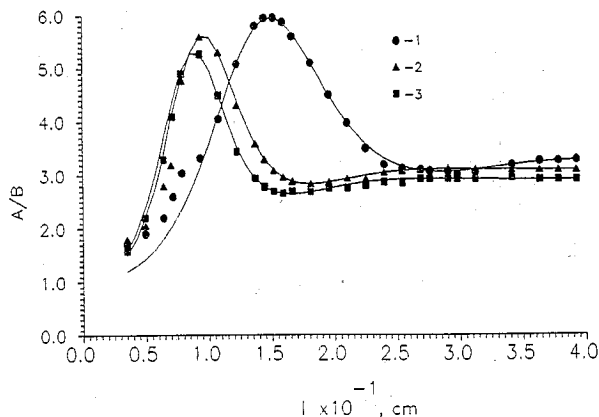


Fig. 5. The experimental (filled symbols) and theoretical (solid lines) values of the line asymmetry parameter A/B of the $C_{15}HNO_3$ plates with dimensions $l \times 0.4 \times 0.02 \text{ cm}^3$ as a function of sample width (l). The experimental curves 1, 2 and 3 correspond to 300, 220 and 183 K, respectively. $B_0 \perp c$, $\nu = 9.52 \text{ GHz}$. The theoretical curves at 300, 220 and 183 K are one-dimensional Dyson's [10] curves with $\sigma_c = 1.0, 2.32$ and $2.57 \Omega^{-1} \text{ cm}^{-1}$, $T_{Da} = 13.1, 11.0$ and 11.5 , in the units of 10^{-8} s , respectively.

from the experimental values of l_{max} and $(A/B)_{\text{plat}}$ and the value of T_{Da} determined by the standard procedure [11, 12] (using the values of ΔB and A/B of GIC plates with l from "plateau" of the $A/B(l)$ dependence). As seen from Fig. 5, the theoretical and experimental dependences $A/B(l)$ are in good agreement at $l > l_{\text{max}}$. Taking it into consideration, the values of σ_c at other temperatures were determined using the experimental values of l_{max} and $(A/B)_{\text{plat}}$ only. By this procedure of σ_c -determination we established that σ_c , as well as σ_a increases with decreasing the temperature (Fig. 3). Hence in $C_{15}HNO_3$, during and after intercalate crystallization the ΔB increases simultaneously with the conductivity in spite of the fact that the g -tensor values of the spin carriers are temperature independent. We have found a possible explanation for this uncommon property of the CESR line width temperature dependence by investigating the temperature effect on N and on the in-plane spin diffusion constant (D_a).

At a given geometry of the GIC sample the CESR signal integral intensity $I = (A + B)\Delta B^2$ is proportional to $N\delta_c$. Because of this, the increase in I simultaneously with the decrease in δ_c , which takes place in $C_{15}HNO_3$ as the temperature decreases from T_{c1} to T_{c3} (Fig. 2), suggests that N increases at this temperature range. This conclusion does not change once the 6.7% increase of I (relative to its value in liquid phase of intercalate) along with the decrease of A/B at this temperature range has been taken into consideration with the help of the corresponding

nomogram in Fig. 4. Further, using the relation $I \sim N\delta_c$ and approximating the experimental values of σ_c before and after phase transition by the linear function $[(-4.77 \times 10^{-3} \text{ K}^{-1}) T + 2.45] (\Omega \text{ cm})^{-1}$ and $[(-1.76 \times 10^{-2} \text{ K}^{-1}) T + 5.79] (\Omega \text{ cm})^{-1}$, respectively, we have calculated the temperature dependence of the relative spin carrier concentration (Fig. 2):

$$N(T)/N(300 \text{ K}) = [I(T)/I(300 \text{ K})] \times [\sigma_c(T)/\sigma_c(300 \text{ K})]^{1/2}, \quad (1)$$

where $N(300 \text{ K})$, $I(300 \text{ K})$ and $\sigma_c(300 \text{ K})$ are the values of N , I and σ_c at 300 K, respectively. As should be expected from the results of the qualitative analysis of the experimental data, N increases as the temperature is lowered in the crystal phase of the intercalate (Fig. 2). Concerning the behaviour of N in the quasi-liquid phase of the intercalate, all its variations occur within the experimental error (Fig. 2). It follows from above that in synthetic metal $C_{15}HNO_3$ in the solid phase of the intercalate the increase of σ_a is at least partially due to the increase in N .

With knowledge of the experimental values of A/B and ΔB of GIC plates with l from "plateau" of the $A/B(l)$ dependence and using the Feher and Kip [11] nomograms one can easily determine the in-plane diffusion time across the skin-depth δ_c (T_{Da}) determined by the σ_c . According to Dyson [10], in approximation of independent electrons, the value of T_{Da} is connected with the value of D_a by the relationship $D_a = \delta_c^2/2T_{Da}$. The estimation of D_a according to these expressions has shown (Fig. 3) that its value during and after intercalate crystallization decreases.

The diffusion constant is equal to

$$D_a = \frac{1}{3} \Lambda_a v_a = \frac{1}{3} \tau_a v_a^2 \quad (2)$$

by definition [3]. In the expression (2) Λ_a , τ_a and v_a are the mean free path, the electroconductivity relaxation time and the spin carriers velocity, all along the basal plane. If in expression (2) v_a is identified with Fermi velocity v_F and it is taken into account that in metals v_F is proportional to the square of current carriers concentration [3] then it may be concluded, from this expression, that τ_a in $C_{15}HNO_3$ decreases during and after intercalate crystallization (since at this process D_a decreases simultaneously with the increase of N). This suggests that "nonmetallic" broadening of the CESR line in $C_{15}HNO_3$ is due to the decrease of the in-plane current carriers mobility.

The broadening of the CESR line is homogeneous. The g -tensor values of spin carriers are independent of the changes in aggregate states of the intercalate and they are close to the g -tensor value of the free electron. These facts testify that the density of charge carriers

on intercalate molecules is small and does not change at phase transition. Above, it is shown that it is possible to analyze the transformations of the electronic properties of $C_{15}HNO_3$ induced by phase transition in terms of the rigid electronic band model of GICs [4, 15]. In terms of this model, in acceptor GICs related to the conductors with the hole Fermi surface [4] the increase of N as the temperature decreases is possible only at lowering of the Fermi energy, for instance, due to the transfer of the additional electrons from carbon layers to intercalate layers [4, 15], or due to the localization of spin carriers in graphite layers, preferentially at the edges of the Dumas–Herold islands [16]. In this case, the increase in N must obviously be accompanied by the decrease in τ_a , and consequently, by the increase in ΔB , because the localized electrons are the perturbation centers for the conduction electrons. That is, in the crystal phase of the “guest” molecules subsystem, the rigid electronic band model predicts qualitatively the correct temperature effect on these experimental values. Note that several recent experimental results confirm the existence in GICs of the localized moments and traps. For instance, Davidov *et al.* [17] observed the presence of localized moments in the graphite layers by spin–echo techniques performed on AsF_5 GICs and suggested the existence of traps. A precise analysis of the spin-relaxation time [13] and the anisotropy of the line width with the orientation of B_0 [18] confirms this assumption. At the present time, the exact nature of this trap remains to be determined.

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